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# Light-induced crystallization of thin Si films

T. Mchedlidze<sup>a</sup>, T. Arguirov<sup>b,c</sup>, and M. Kittler<sup>b,c</sup>

<sup>a</sup>Institut für Angewandte Physik, TU Dresden, Germany

<sup>b</sup>Joint Laboratory IHP/BTU, Cottbus, Germany

<sup>c</sup>IHP, Frankfurt (Oder) Germany



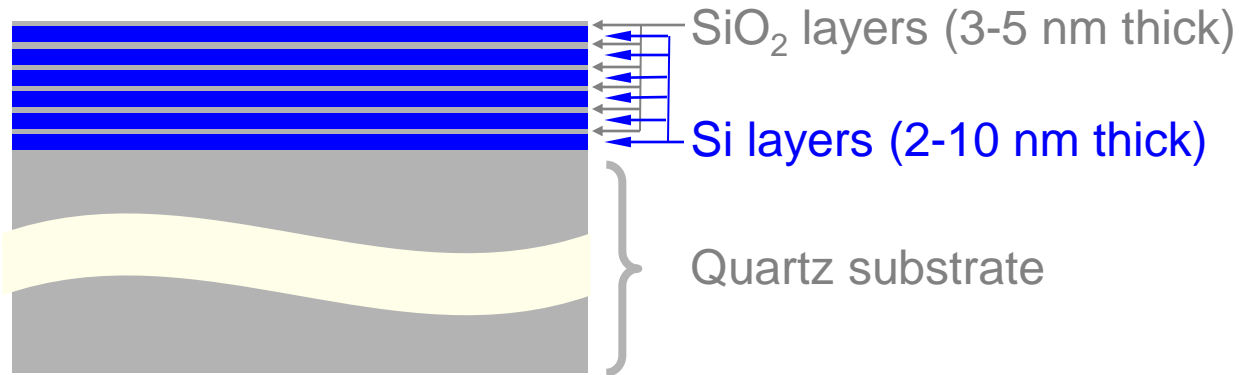
# Outline

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- Motivation
- Experiments with light-induced crystallization
- Model, calculations, results and discussion
- Summary



High crystal quality Si/SiO<sub>2</sub> MQW on cheap substrate:



### Fabrication:

- Subsequent deposition of a-Si and SiO<sub>2</sub> layers on a quartz substrate;
- Crystallization of Si layers.

### Possible future applications:

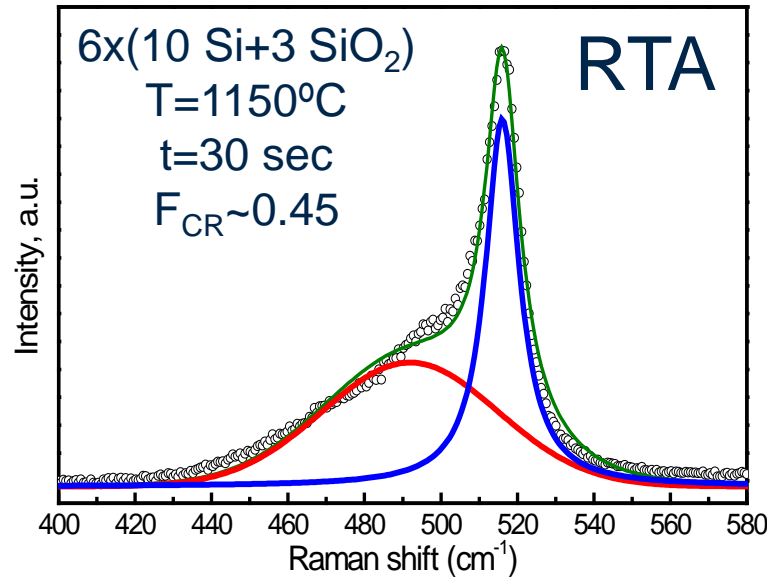
- Tandem solar cells
- 3<sup>rd</sup> generation PV materials
- Photonics
- Device apps, nanoelectronics
- ...

High crystal quality of Si-nc layers is crucial!

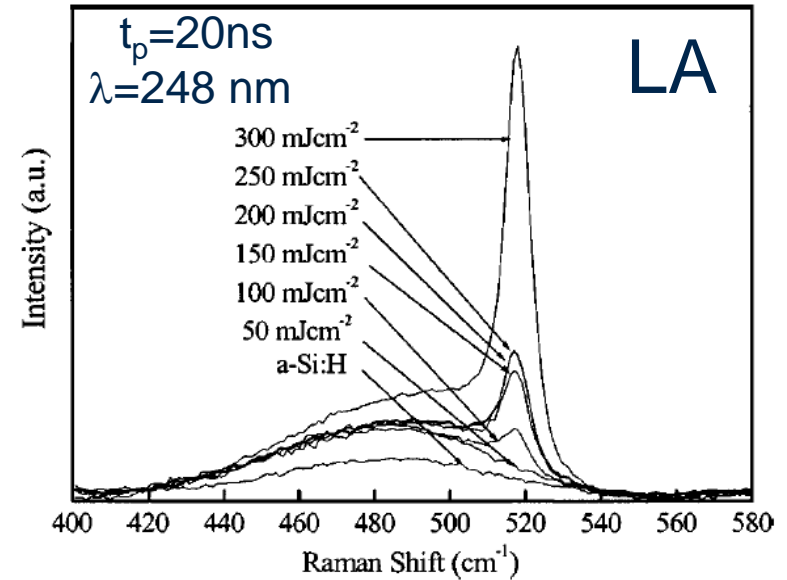


# Problems with crystallization of Si/SiO<sub>2</sub> MQW

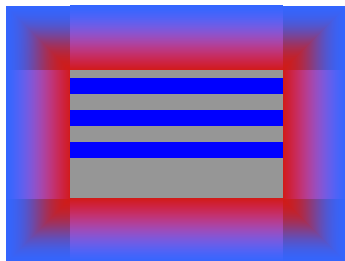
[T. Mchedlidze, et al.  
Thin Solid Films **516**, 6800 (2008)]



[A. Adikaari and S. Silva  
J. Appl. Phys. **97**, 114305 (2005)]



- $F_{CR} = I(\text{Si}_{CR}) / I(\text{Si}_{TOT}) < 0.7$
- High residual compressive stress



## “Bulk” heating:

High level of crystallization (>80%) *impossible*.

Reasons: Si melting, mismatch in the thermo-physical properties of MQW components.



# Smart light-induced crystallization

## What to do:



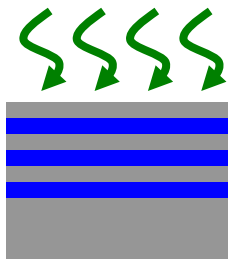
- Preferential heating of a-Si layers;
- Avoid a-Si melting ► solid phase transition;
- Input proper amount of energy.

## Concept:

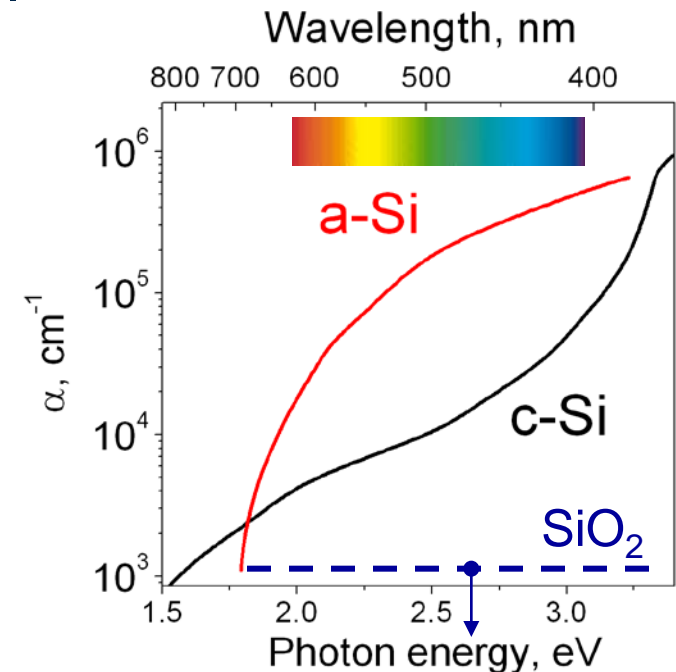
Employ differences in light absorption between a-Si, Si-nc, SiO<sub>2</sub> and a substrate;

- Self-regulated crystallization process.

## Realization:



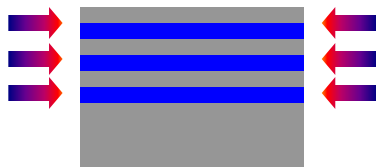
Convert a-Si to Si-nc using proper light flux @ ~500 nm.





# Smart light-induced crystallization

## What to do:



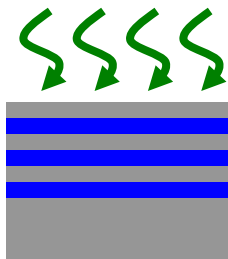
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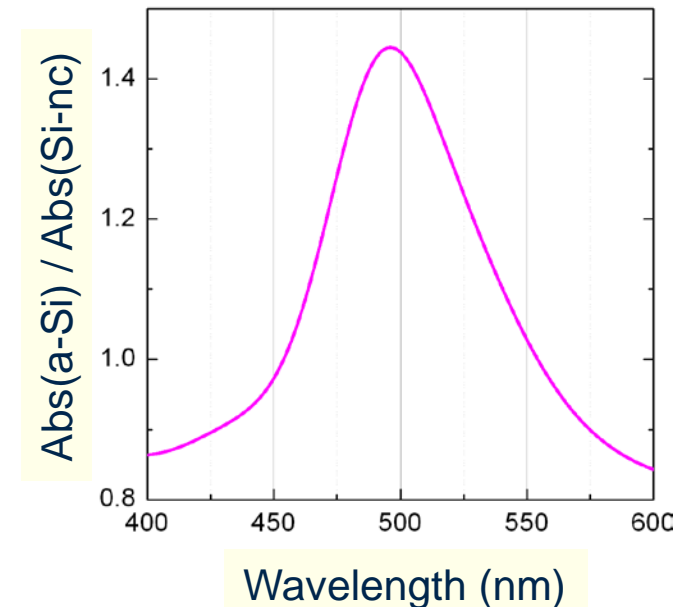
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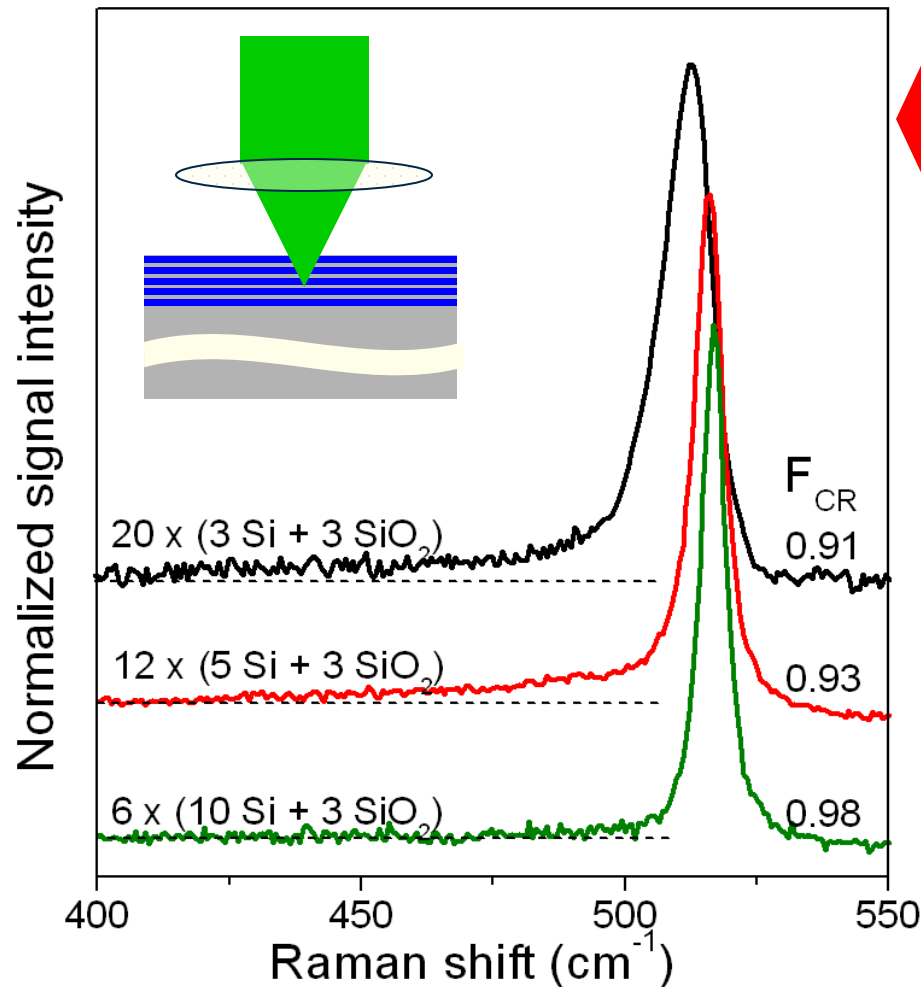


Convert a-Si to Si-nc using proper light flux @ ~500 nm.



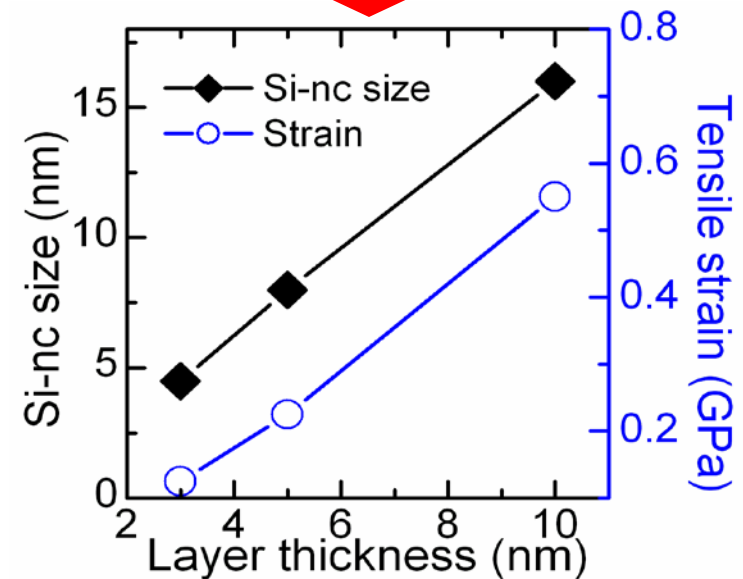


# Light induced crystallization: static mode



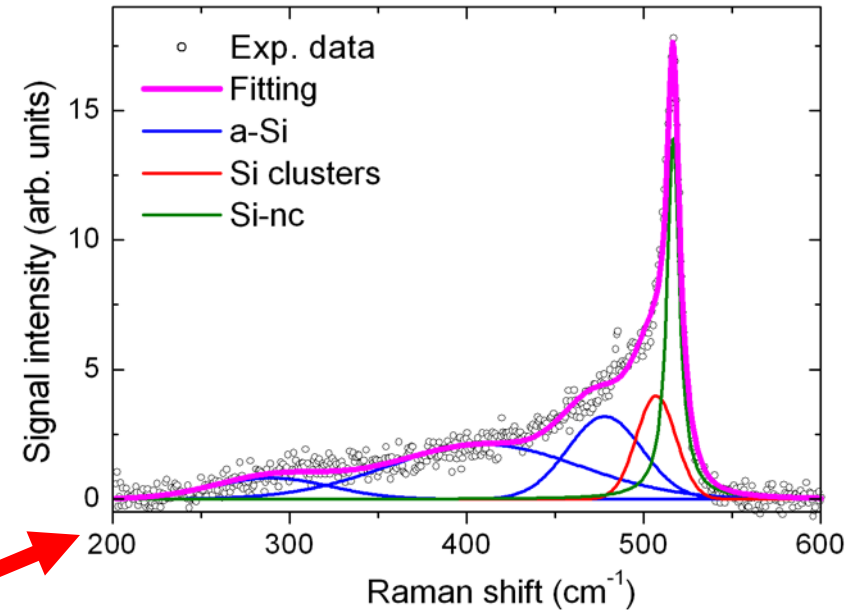
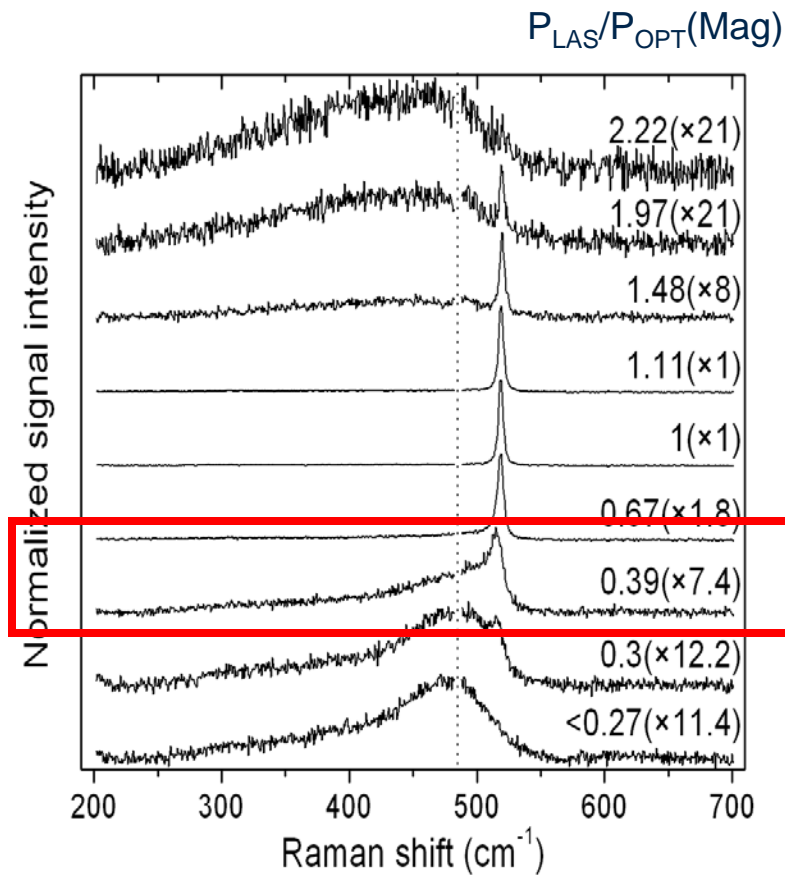
Raman spectra detected from various MQW crystallized in nearly optimal conditions of light flux @ 532 nm.

Sizes of Si-nc crystallites and strain in the MQWs estimated from Si-nc peak characteristics.



[T. Mchedlidze, et al., Phys. Rev. B **77**, 161304(R) (2008)]

# Influence of light flux value



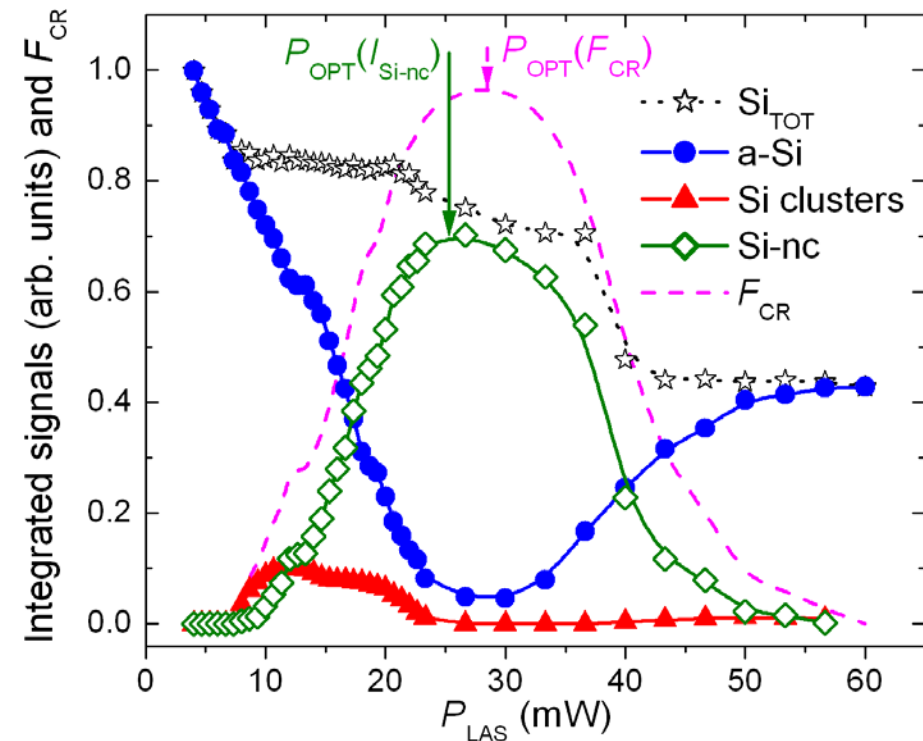
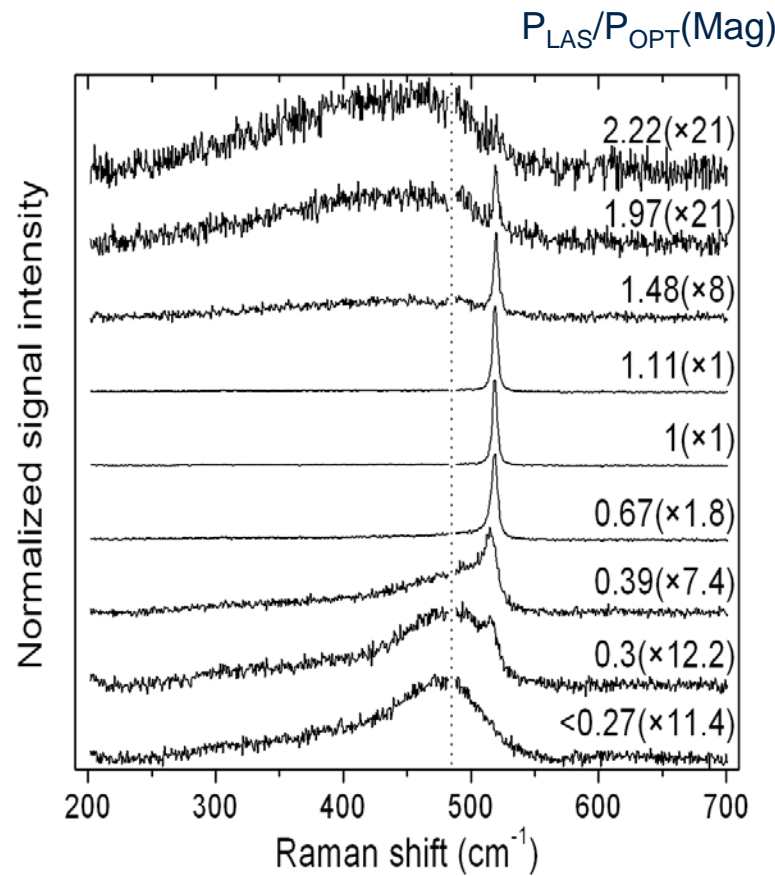
$I_{\text{Si-nc}}$

$$F_{\text{CR}} = I_{\text{Si-nc}} / I_{\text{Si}}$$

[T. Mchedlidze, et al., J. Appl. Phys. 107, 124302 (2010)]



# Influence of light flux value



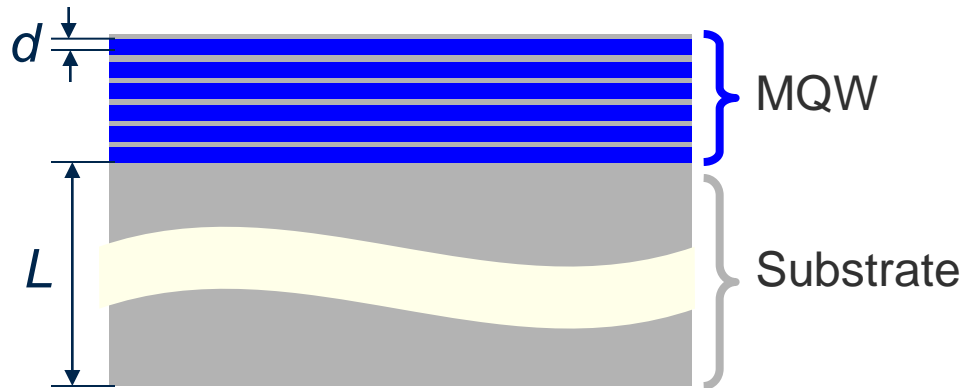
$$\eta = \left[ \frac{I_{Si-nc}}{P_{LAS}} \right]$$

$$PR = \frac{P_{OPT}(I_{Si-nc})}{P_{OPT}(F_{CR})}$$

[T. Mchedlidze, et al., J. Appl. Phys. 107, 124302 (2010)]



# Heating of MQW by light: assumptions



**(a) Light is absorbed “only” in the Si layers of MQW**

since @  $\lambda_{\text{LAS}} = 532 \text{ nm}$   
 $\alpha_{\text{QUARTZ}} \ll \alpha_{\text{Si-nc}} < \alpha_{\text{a-Si}}$

**(b) Heat transfer is determined by the properties of the substrate**

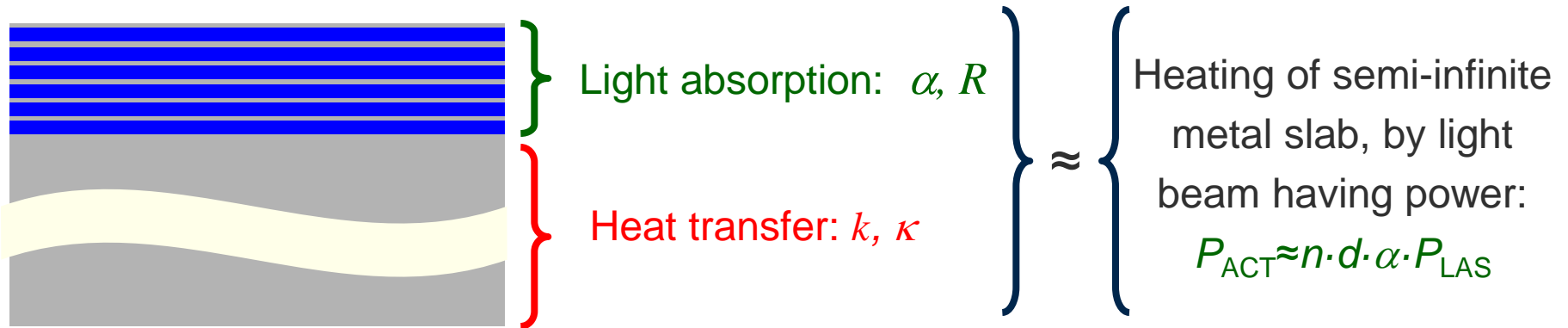
due to  $n \cdot d \ll L$  and negligible lateral heat transfer in MQW layers

**(c) consider heat transfer only perpendicular to the MQW surface**

since beam radius  $\gg n \cdot d$

**(d) Optical interference** in MQW layers and **temperature dependences of the material properties could be neglected.**

# Heating of MQW by light: modeling



$$\Delta T(\tau) = \frac{\alpha P_{ACT} \beta^2 \kappa (1 - R)}{\pi k} \int_0^\tau \frac{\exp(\alpha^2 \kappa t) \operatorname{erfc}(\alpha \sqrt{\kappa t})}{1 + 4 \beta^2 \kappa t} dt$$

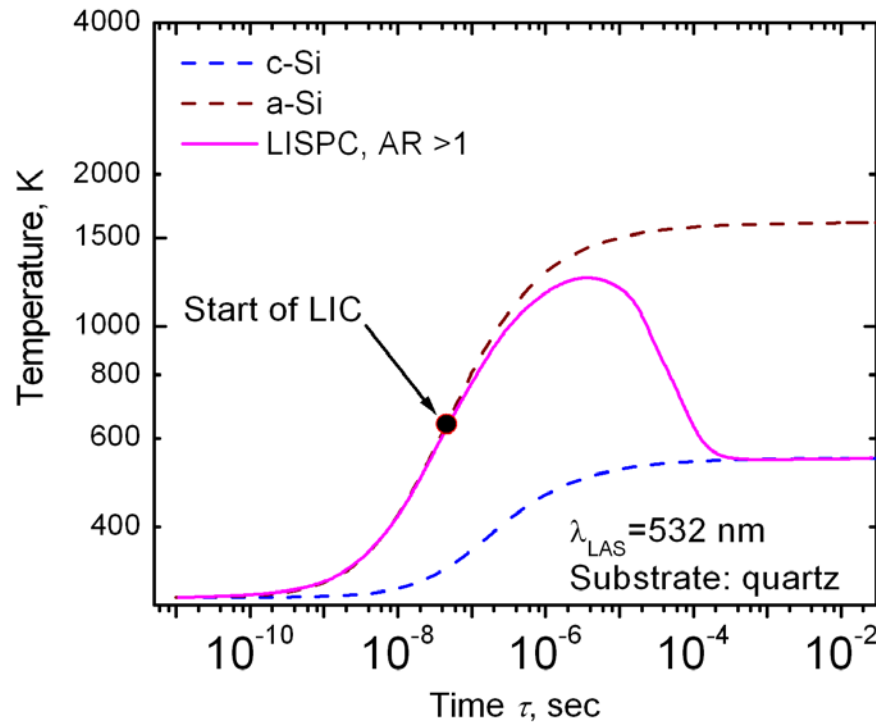
[B. J. Bartholomeusz, J. Appl. Phys. **73**, 1066 (1993)]

$\alpha$  is absorption coefficient,  $k$  is thermal conductivity,  $\kappa$  is thermal diffusivity,  $R$  is reflectance,  $\beta = 1 / \sqrt{2} \sigma_{eff}$ , with  $\sigma_{eff}$  an irradiance radius of the laser spot. All parameters could be found in publications.

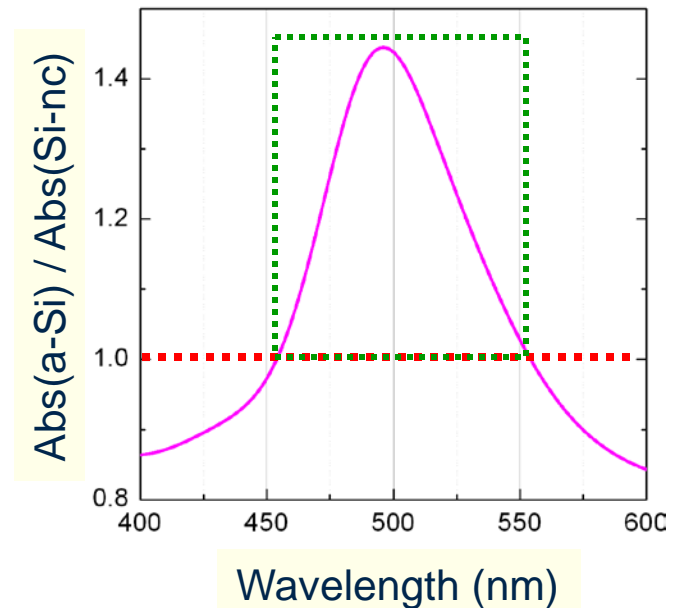
# Heating of MQW by light + LIC



Optimal light flux,  $\lambda=532$  nm



$$AR(\lambda) = \frac{Abs(a - Si)}{Abs(Si - nc)}$$



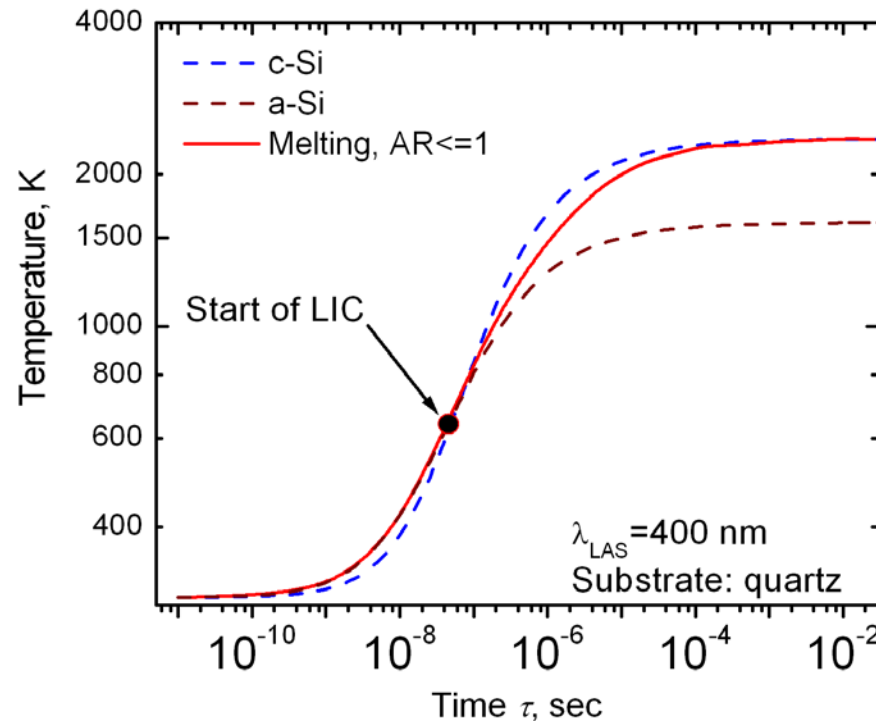
**Wavelength dependence of LISPC was obtained experimentally**

[T. Mchedlidze, et al., J. Appl. Phys. 107, 124302 (2010)]

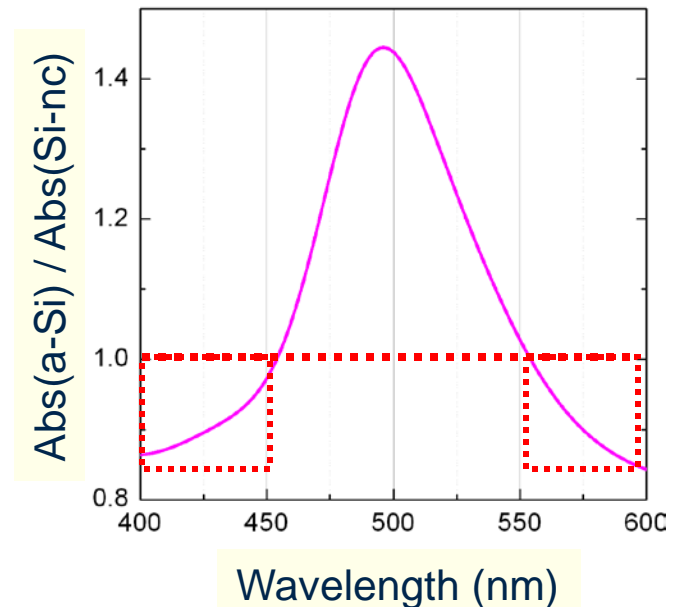
# Heating of MQW by light + LIC



Optimal light flux,  $\lambda=400$  nm



$$AR(\lambda) = \frac{Abs(a-Si)}{Abs(Si-nc)}$$

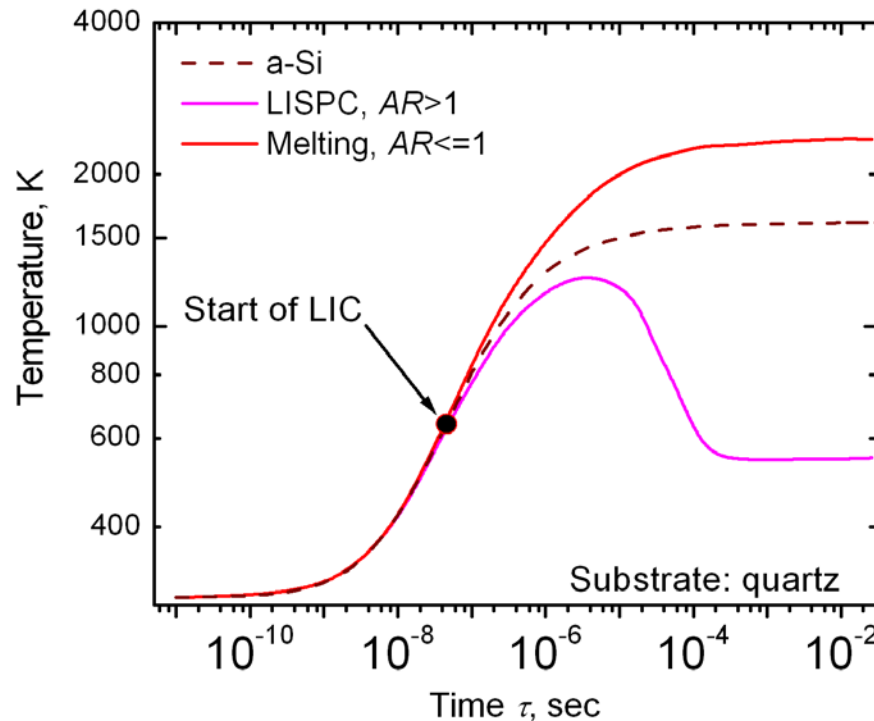


**Wavelength dependence of LISPC was obtained experimentally**

[T. Mchedlidze, et al., J. Appl. Phys. 107, 124302 (2010)]



# Heating of MQW by light + LIC



$$AR(\lambda) = \frac{Abs(a - Si)}{Abs(Si - nc)}$$

- For  $AR > 1$ , “negative feedback”, self-regulated LISPC possible.
- For  $AR \leq 1$ , “positive feedback”, LIC easily transfers to melting.
- Larger  $AR \rightarrow$  more efficient LISPC, less changes in MQW after finish of crystallization.

**Wavelength dependence of LISPC was obtained experimentally**

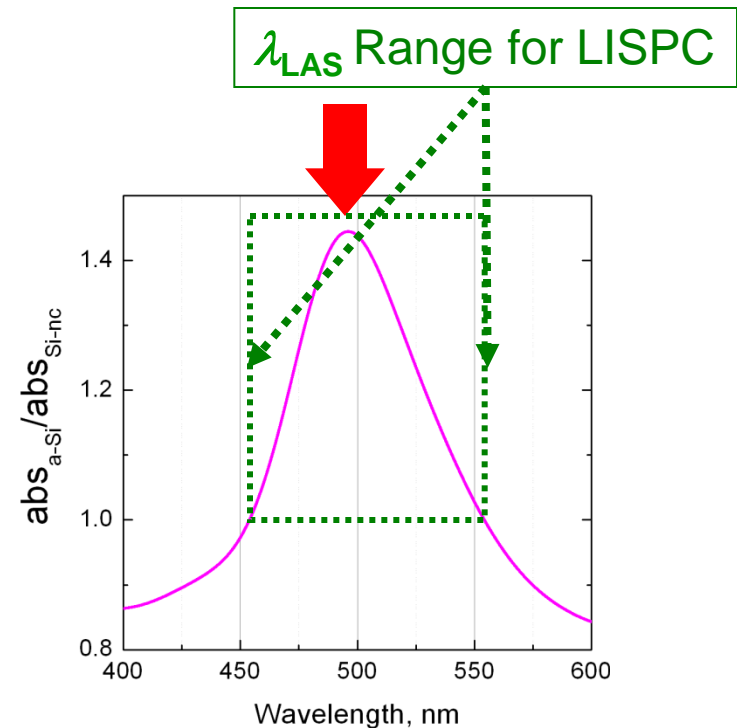
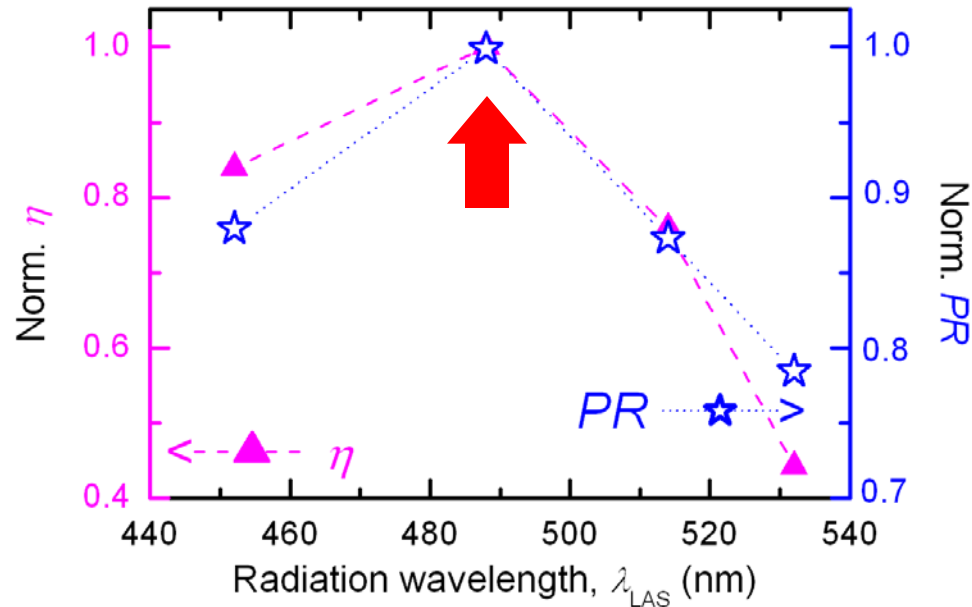
[T. Mchedlidze, et al., J. Appl. Phys. 107, 124302 (2010)]

# Efficiency of LIC at various $\lambda_{LAS}$



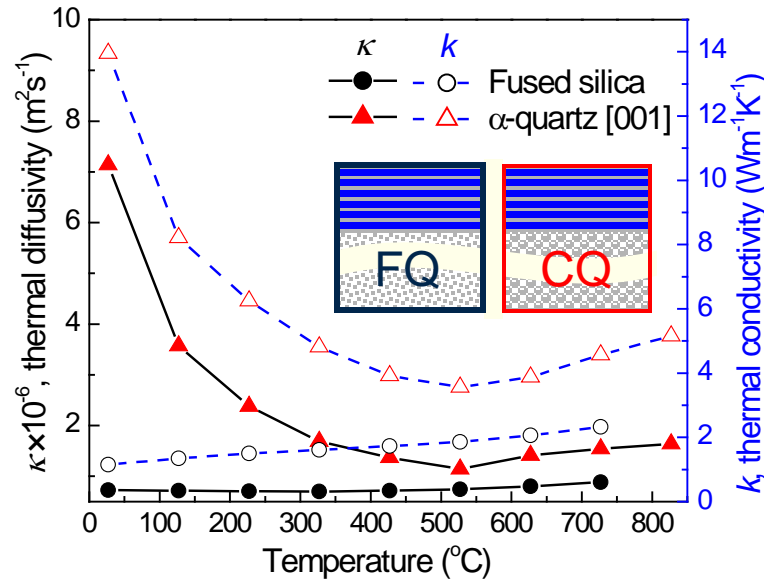
$$\eta = \left[ \frac{I_{Si-nc}}{P_{LAS}} \right]_{OPT} \rightarrow \eta(\lambda_{LAS}) \propto \frac{I_{Si-nc}(\max)}{P_{OPT} \cdot [1 - \exp(-\alpha nd)]}$$

Optimal power ratio  $\rightarrow PR(\lambda_{LAS}) = \frac{P_{OPT}(I_{Si-nc})}{P_{OPT}(F_{CR})}$





# Efficiency of LISPC for various substrates

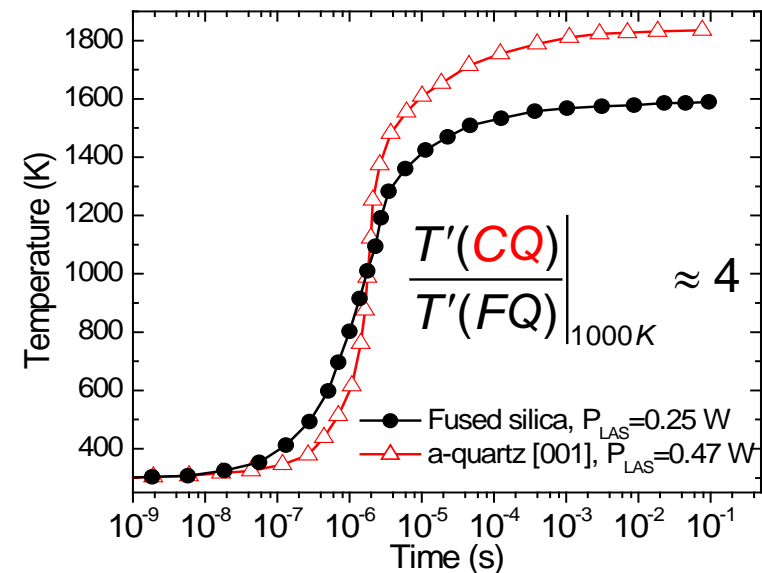


- $T(t)$  dependencies could be roughly estimated using  $\kappa(T)$  and  $k(T)$ .
- Larger  $T'=(dT/dt)$  at the moment of crystallization for  $\alpha$ -quartz substrate.

@ optimal LISPC conditions:

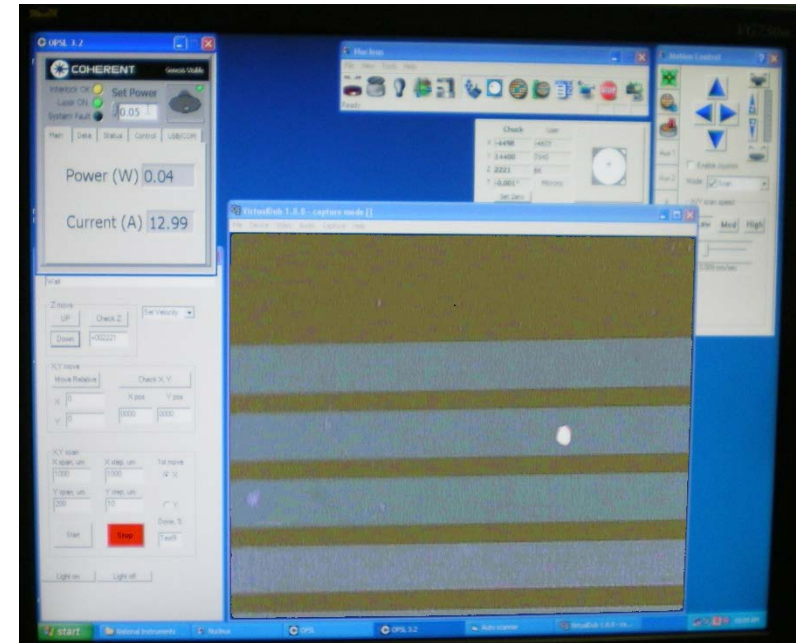
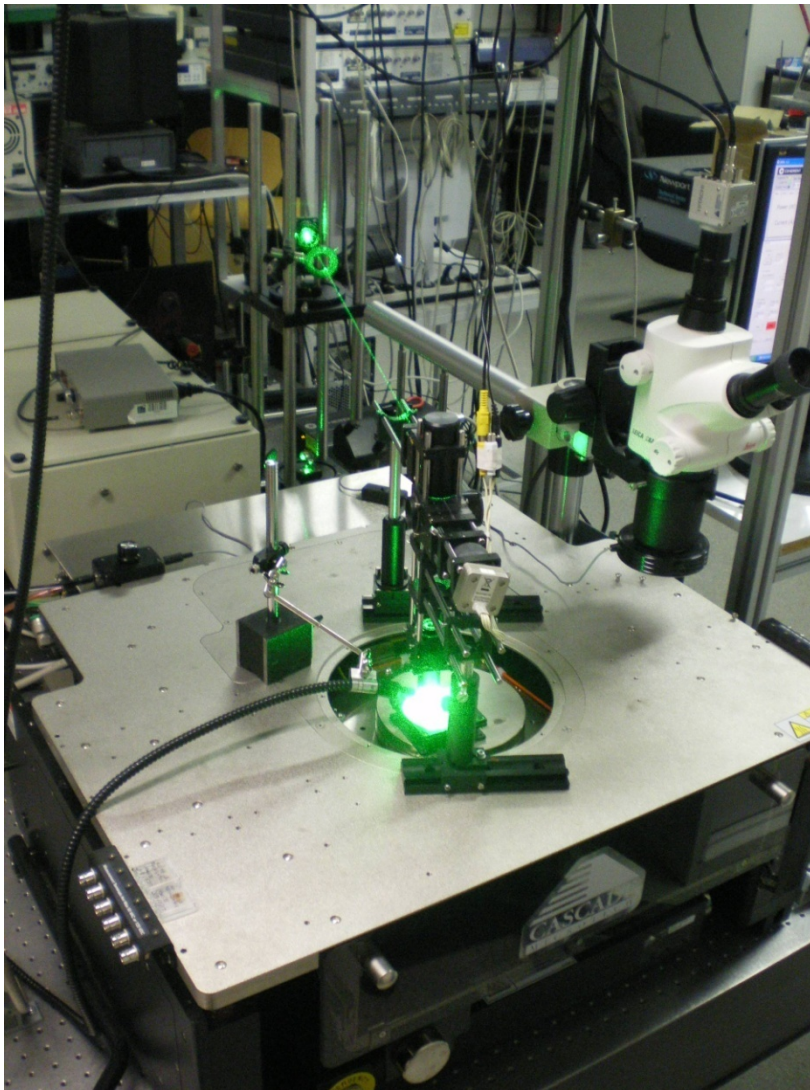
- $I_{\text{Si-nc}}(\text{FQ}) \approx 1.25 I_{\text{Si-nc}}(\text{CQ})$
- $P_{\text{OPT}}(\text{FQ}) \approx 0.5 P_{\text{OPT}}(\text{CQ})$
- Large compressive (CQ) vs. small tensile (FQ) residual stress.

$$\frac{\eta(\text{FQ})}{\eta(\text{CQ})} \approx 2.5$$





# Installation for fast LISPC


$$\lambda_{LAS} = 532 \text{ nm}$$

Laser power ~ 400 mW

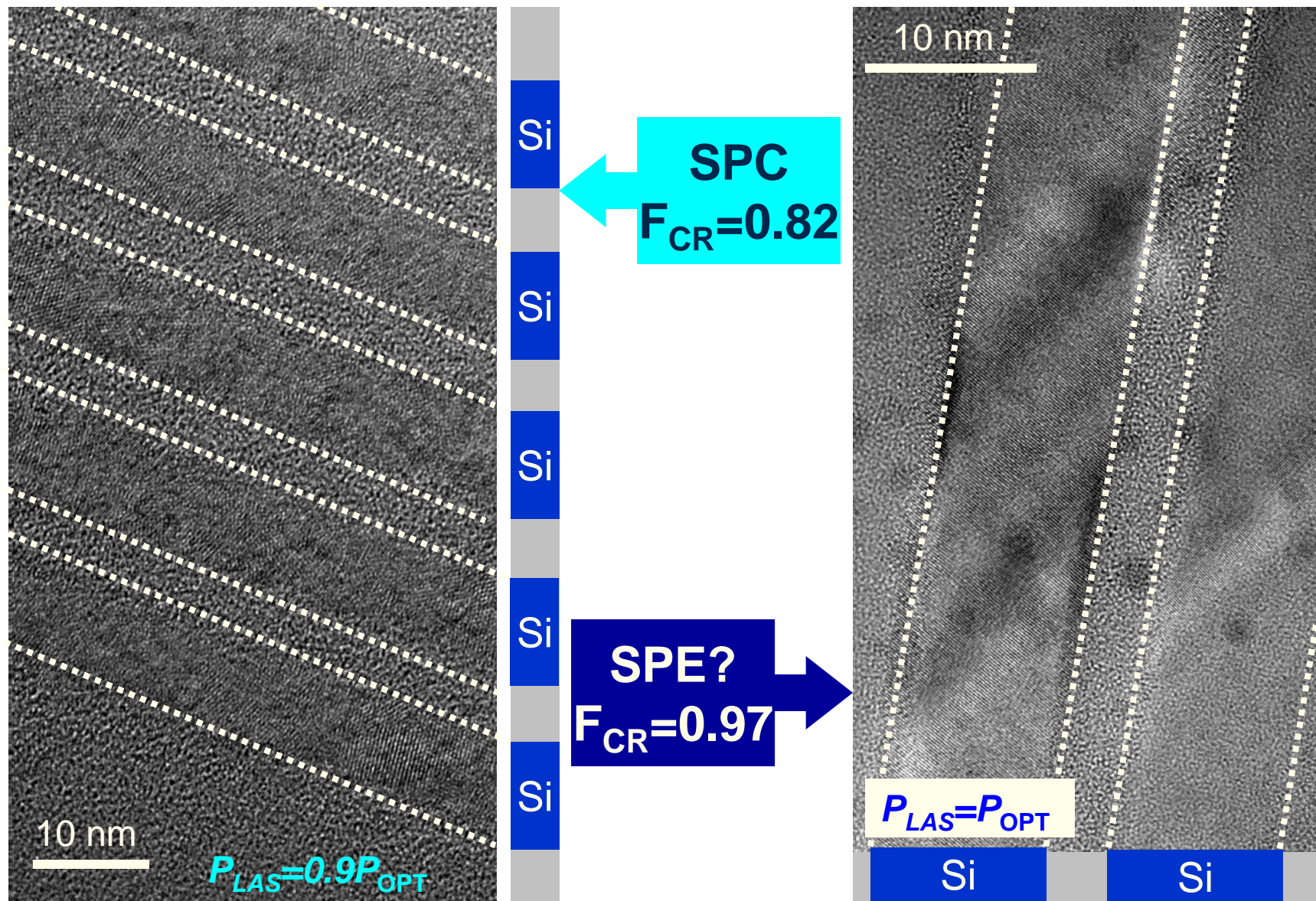
Focal distance : 12-15 mm

Scanning speed: 0.3 mm/sec

Process rate:  $5 \times 10^{-3} \text{ mm}^2/\text{sec}$

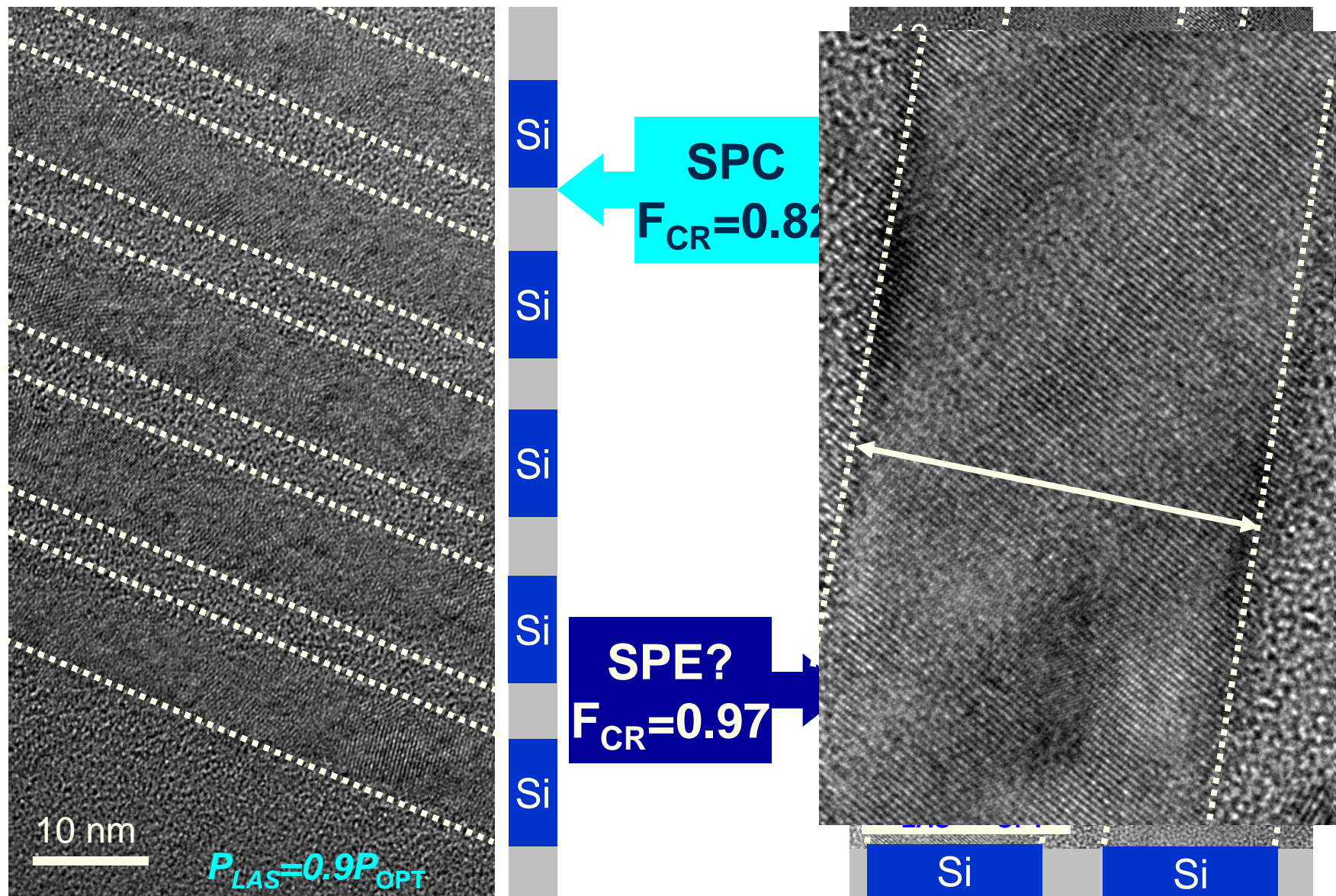


# XHRTEM for MWQ with 10 nm thick Si layers





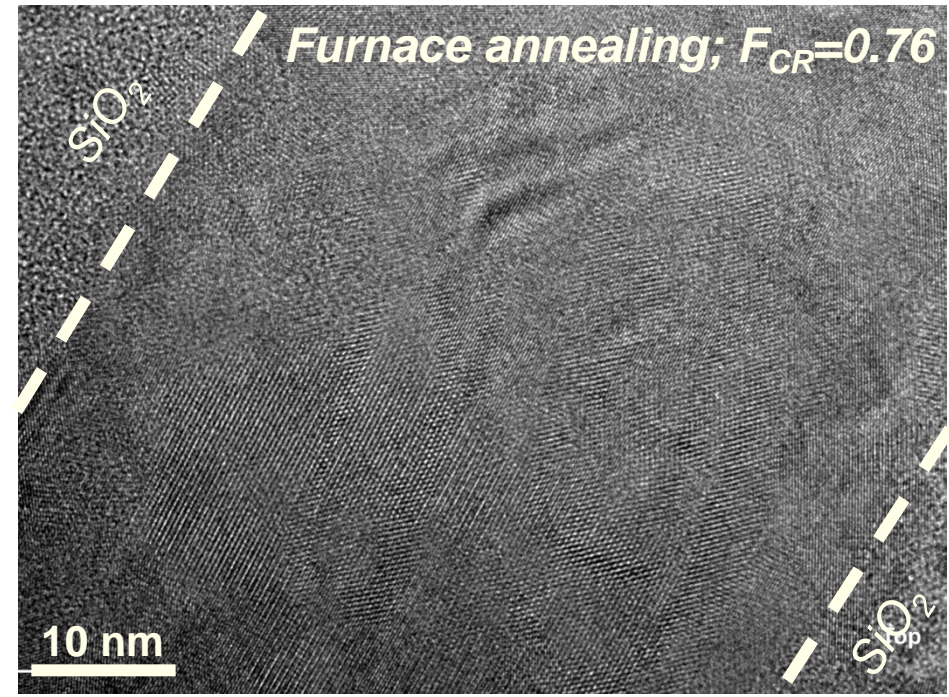
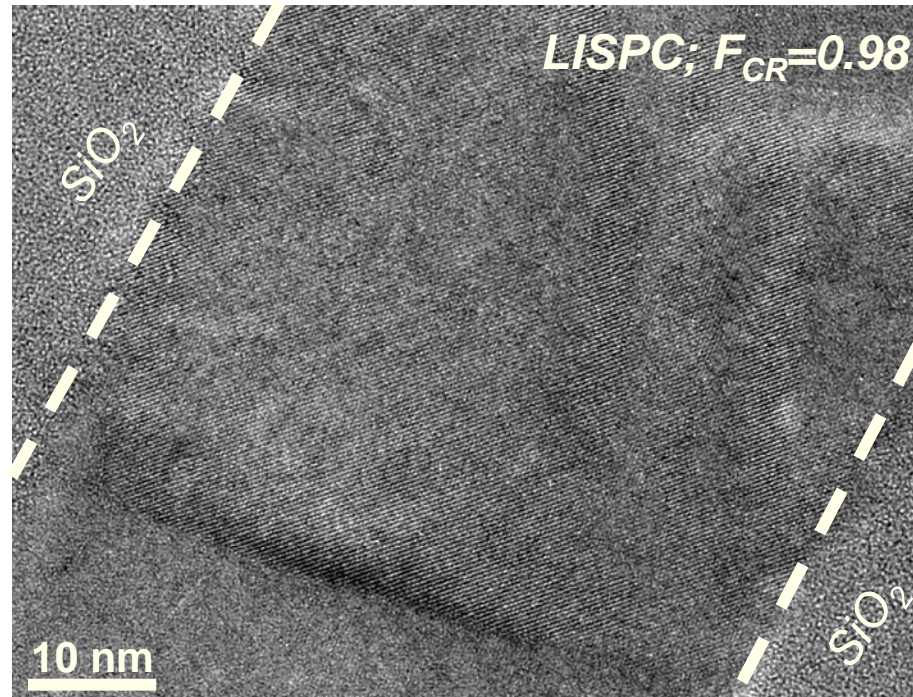
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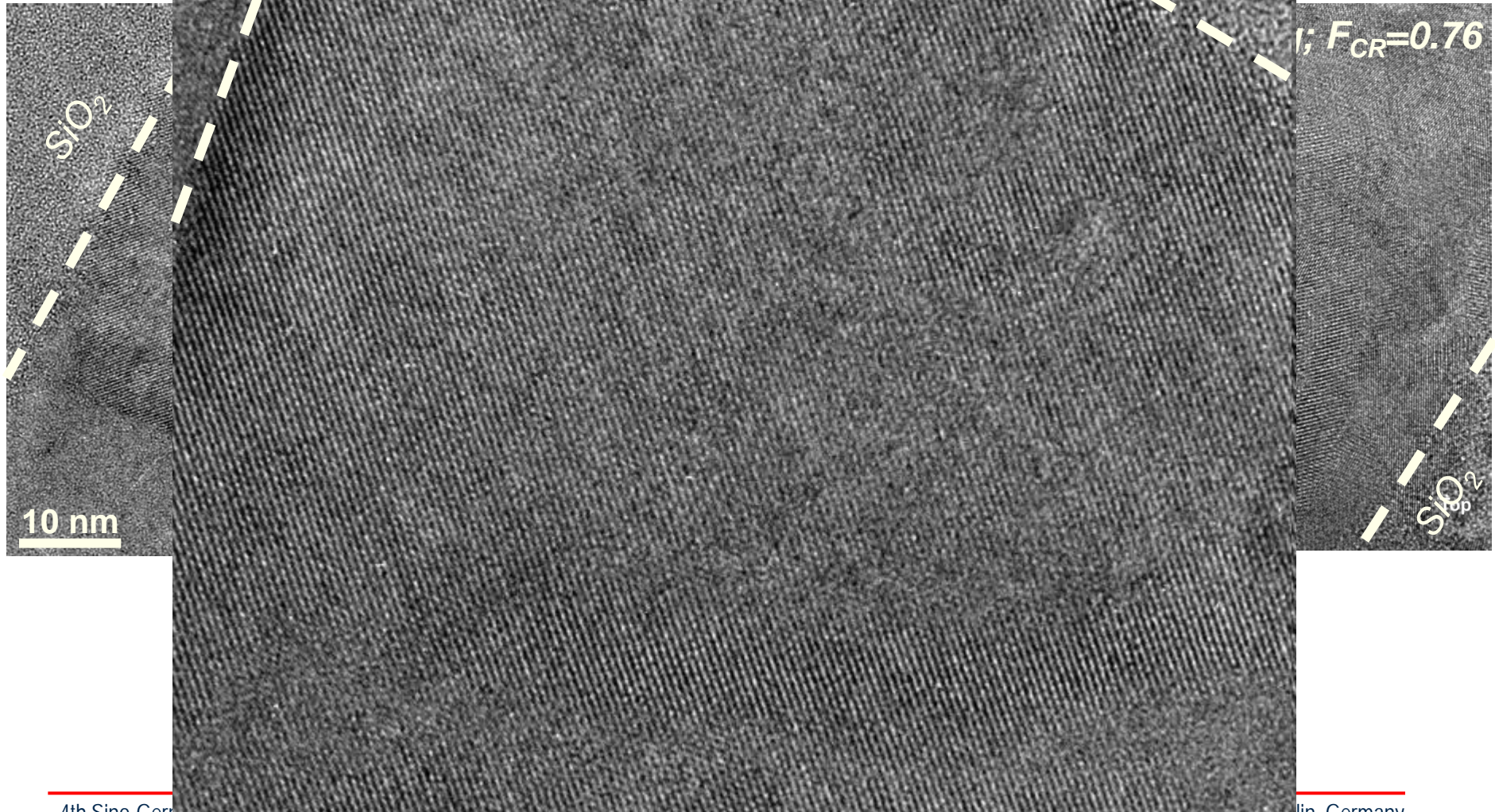


# XHRTEM for 60 nm Si film inside SiO<sub>2</sub> layers



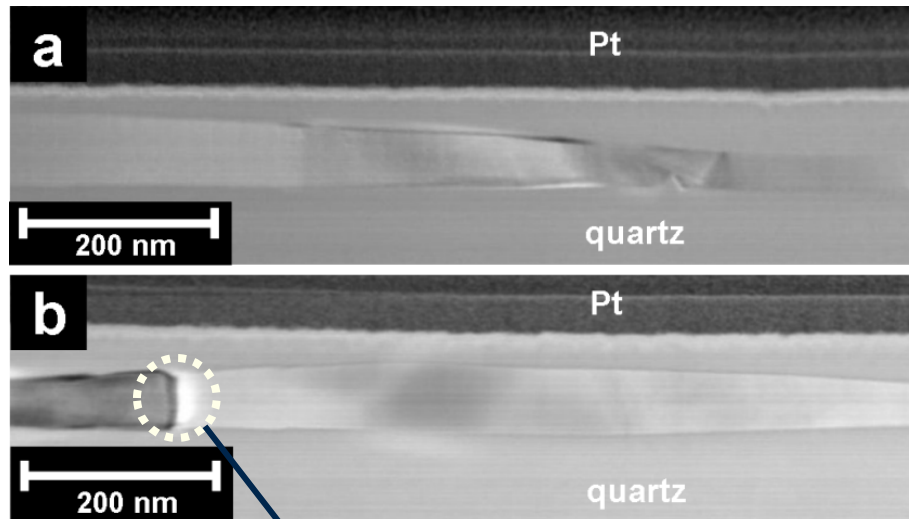


XHR

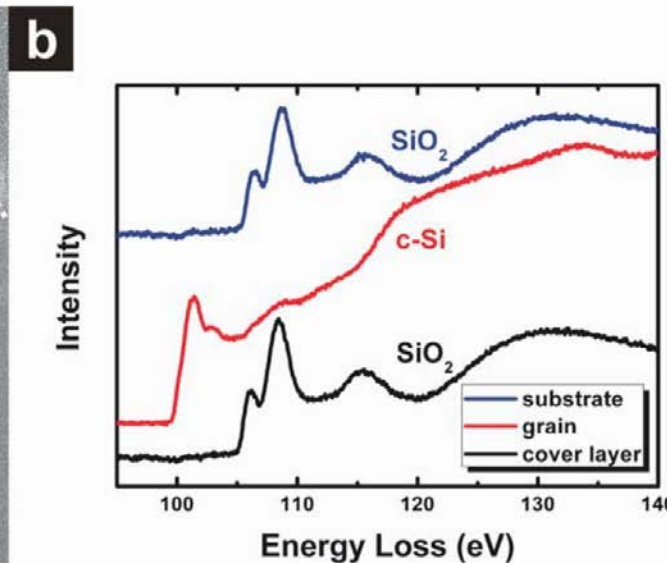
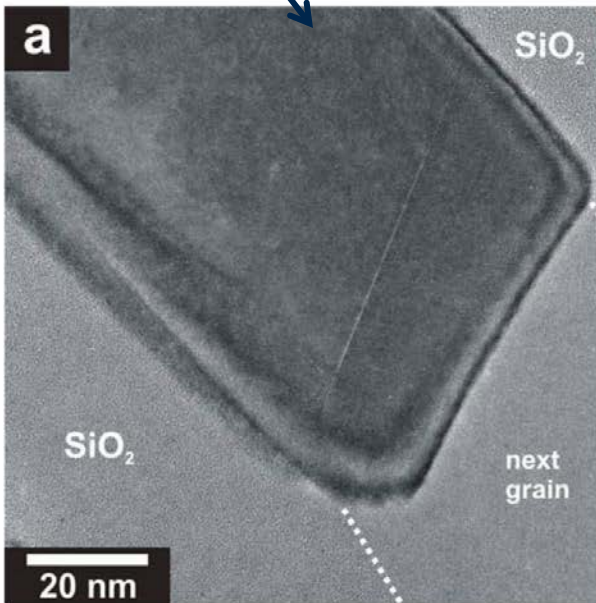




# Quality for 60 nm Si film inside $\text{SiO}_2$ layers



Long-range “panoramic” images show high quality crystalline layers for the distances comparable with the diameter of light beam.



The EEL spectrum of the grain is typical for single crystalline silicon with a low content of lattice defects and reflects the high grade of crystallinity.



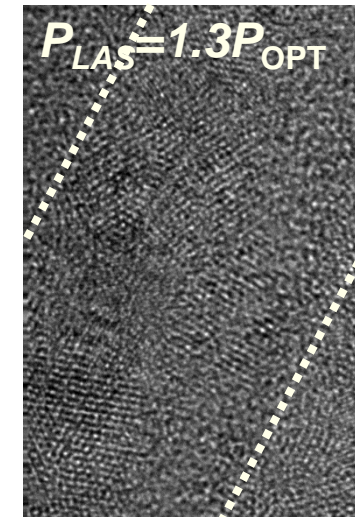
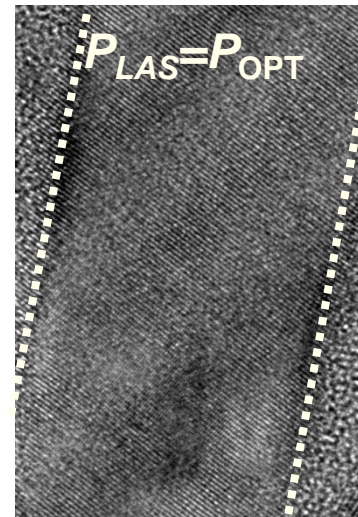
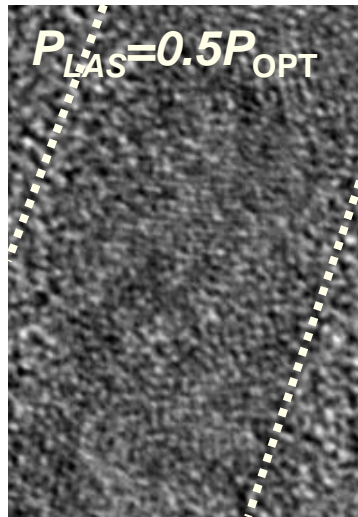


# Influence of hydrogen and external conditions

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- **Hydrogen** contained in a-Si layers enhances possibilities of successful LIC. For Si layers with outdiffused hydrogen “window” of optimal illumination conditions shrinks. However, too much hydrogen is also not good!
- Presence of **oxygen** in the ambient has small influence on the process – “internal” oxidation plays the major role.
- **Thermal properties of ambient** can change thermal exchange in the system during crystallization. Namely, LISPC in water ambient although possible requires much larger light fluxes.
- **Gas pressure** has small influence on the crystallization process itself but at low ambient pressure outgassing of hydrogen may cause damage of the MQW structure (formation of craters).
- The results on controlling of **substrate temperature** can be applied for optimization of LISPC process for various substrate materials.

# Summary: Light power dependence



$$(0.95 \div 1.05) P_{\text{OPT}}$$

$$F_{\text{CR}} = 0.95 \div 1$$

**SPE**

$$(0.8 \div 0.95) P_{\text{OPT}}$$

$$F_{\text{CR}} = 0.8 \div 0.95$$

**SPC**

$$(1.1 \div 1.5) P_{\text{OPT}}$$

$$F_{\text{CR}} = 0.9 \div 0.5$$

**Melting**

$$(0.3 \div 0.8) P_{\text{OPT}}$$

$$F_{\text{CR}} = 0.1 \div 0.8$$

**Nucleation**

$$\text{Flux} = n \times 10^5 \text{ W/cm}^2$$

$$(1.5 \div 2) P_{\text{OPT}}$$

$$F_{\text{CR}} = 0.5 \div 0.1$$

**Intermixing**

$$< 0.3 P_{\text{OPT}}$$

$$F_{\text{CR}} < 0.1$$

**Heating**

$n$  depends on:  $\lambda$ , H content,  
substrate, layer thickness...

$$> 2 P_{\text{OPT}}$$

$$F_{\text{CR}} < 0.1$$

**Dissolution**





# Summary

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- LISPC is fast, self-regulated, a-Si→Si-nc solid-to-solid transition process at  $\lambda$  when  $\alpha_{\text{a-Si}}(\lambda) > \alpha_{\text{Si-nc}}(\lambda)$ ;
- The maximal efficiency for LISPC:
  - $\alpha_{\text{a-Si}}(\lambda) / \alpha_{\text{Si-nc}}(\lambda)$  @ max;
  - Light flux should be maximally accommodated by the crystallization process;
  - Optimal substrate should be used.
- Crystalline quality after LISPC is much better than after RTA or LA.
- The process should be applicable also for crystallization of a-Si films on a substrate.



# Acknowledgements

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- For deposition of MQWs the authors would like to thank **B. Berghoff and N. Wilck** (RWTH, Aachen, Germany).
- For TEM analyses the authors would like to thank **M. Beigmohamadi and J. Mayer** (Forschungszentrum, Jülich, Germany)  
**M. Schade and H. S. Leipner** (Interdisziplinäres Zentrum für Materialwissenschaften, Halle, Germany).
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Thank You for attention!